Investigation of electronic and optical properties of wurtzite MgZnO using GGA + U formalism

R. Ibrahem, P. Narin, S. B. Lisesivdin & E. Ozbay

To cite this article: R. Ibrahem, P. Narin, S. B. Lisesivdin & E. Ozbay (2019) Investigation of electronic and optical properties of wurtzite MgZnO using GGA + U formalism, Philosophical Magazine Letters, 99:11, 424-433, DOI: 10.1080/09500839.2019.1696998

To link to this article: https://doi.org/10.1080/09500839.2019.1696998

Published online: 04 Dec 2019.
Investigation of electronic and optical properties of wurtzite MgZnO using GGA + U formalism

R. Ibrahem, P. Narin, S. B. Lisesivdin and E. Ozbay

ABSTRACT

In this study, the electronic and optical properties of wurtzite Mg$_x$Zn$_{1-x}$O structures for different Mg mole fractions ($x$) are studied using Density Functional Theory (DFT). In calculations, the generalised gradient approximation (GGA + U) formalism is used with the Hubbard parameters ($U$) are applied to Zn-3d and O-2p electrons of ZnO. The calculated electronic band structures show that the band gap energies of the investigated structures increase linearly with increasing Mg mole fraction from 0 to 31.25% which is also quantitatively consistent with the previous experimental results. In addition, the electron effective masses of investigated Mg$_x$Zn$_{1-x}$O structures are calculated. The electron effective masses of investigated structures show an increment linearly with increasing Mg mole fractions. The optical results show that the absorption edges of the structures move toward the higher energies region as the Mg mole fractions increase.

1. Introduction

II–VI semiconductors are one of the most important material groups that caused the technological revolution. Zinc oxide (ZnO), which is one of II–VI semiconductors took a good place from both experimental and theoretical investigations because of its electronic, optical and piezoelectric properties such as a direct and wide band gap (3.37 eV), a large exciton binding energy ~60 meV at room temperature, a large piezoelectric coefficient, a strong luminescence, a high thermal conductivity, a high transparency in the visible region and radiation hardness [1–7].

When ZnO is alloyed with MgO, an Mg$_x$Zn$_{1-x}$O structure can be obtained which has a stable wurtzite crystal structure up to 30–36% of Mg mole fraction.
Mg$_x$Zn$_{1-x}$O material can be a useful semiconductor because it has a large band gap (3.3–7.8 eV) \[8,9\]. This alloy has unique electronic and optical properties and these properties make Mg$_x$Zn$_{1-x}$O a very suitable material for producing optoelectronic devices that are working at short wavelengths.

In recent decades, a lot of theoretical investigations that are relied on Density Functional Theory (DFT) had studied the electronic and optical properties of ZnO \[10–12\]. In these investigations, the local-density approximation (LDA) and generalised gradient approximation (GGA) functionalism have been used. It is known that the LDA and GGA functionals are insufficient to explain accurately the electronic properties of ZnO \[13\]. These functionals are underestimated the electronic band gap energy of ZnO, misplaced the energy levels for the Zinc (Zn)-3d states and overestimated the crystal-field splitting energy \[14\].

In these formalisms, the strong hybridisation of localised Zn-3d electrons with Oxygen (O)-2p electrons are ignored. Zn-3d electrons localise at very low binding energies that drive to strong hybridisation with an O-2p electron, this leads to a reduction of the band gap of ZnO \[15\]. This hybridisation is very important for the band gap formation in ZnO, therefore the band gap of ZnO relies on the Zn-3d and O-2p orbitals \[16\]. Recently, some of the theoretical investigations have studied the effects of Hubbard Parameters ($U$) on p orbitals of O and d orbitals of transition metals \[17–19\]. As a result of that, the correction was applied to the d orbitals of Zn and p orbitals of O \[20\]. In these methods, where they are called as GGA + $U$ or LDA + $U$, the orbital-dependent term has been added to the exchange–correlation potential.

In addition, there are few theoretical studies that relied on DFT shed light on the effect of Mg on the electrical and optical properties of ZnO, in which the LDA and GGA functionals had been used to investigate the properties of wurtzite-MgZnO \[21, 22\]. The results showed the band gap linearly increased when the mole fraction of magnesium increased but the band gap values still have underestimation comparing with the experimental studies \[8\]. Few of theoretical studies have been used GGA + $U$ to study Mg effects on wurtzite-ZnO properties and the results were very close to the experimental results \[20\].

In this work, the electronic and optical properties of wurtzite Mg$_x$Zn$_{1-x}$O for different Mg mole fractions (x) from 0% to 31.25% have studied using DFT where the geometry of the structure is optimised with an analytical potential and lattice parameters are taken from experimental literature. The calculations have performed using the GGA + $U$ formalism. The electronic band structures, the density of states (DOS), the electron effective masses and the absorption spectrum of studied Mg$_x$Zn$_{1-x}$O structures have calculated.

### 2. Calculation method

The calculations were performed using the Atomistix Toolkit-Virtual Nano Lab (ATK-VNL) software based on DFT. The calculations were carried out using
GGA + $U$ functionalism. $U$ parameters were applied to Zn-3d electrons and O-2p electrons of Mg$_x$Zn$_{1-x}$O to describe the on-site Coulomb corrections. The best $U$ parameters for Zn-3d and O-2p orbitals of Mg$_x$Zn$_{1-x}$O were used as $U_d = 10$ eV and $U_p = 7$ eV, respectively [23]. For 2×2×2 super cell which has 32 atoms, the mesh cut-off energy and k-points were used as 500 eV and 6×6×5, respectively. Figure 1 shows the crystal structure of the 2×2×2 supercell of the pure ZnO. For each Mg$_x$Zn$_{1-x}$O structure, Mg positions were randomly inserted into Zn lattice points. Random Mg atom relocation is done for ten different random location choices and in every calculation results of electronic band structure show high similarity (i.e. bandgap values change between 1% and 2% with respect to each other). After building a randomly arranged Mg$_x$Zn$_{1-x}$O super cell, the structures were geometrically optimised under Perdone’s analytic potential [24]. In geometric optimizations, the experimental lattice parameters of Mg$_x$Zn$_{1-x}$O crystal were used as $a = b = 0.32491 + 0.047x$ nm and $c = 0.52042 − 0.072x$ nm [25]. The electronic and the optical calculations of Mg$_x$Zn$_{1-x}$O were performed for different mole fractions of Mg (0%, 6.25%, 12.5%, 18.75%, 25% and 31.25%), respectively.

3. Results and discussion

The electronic properties such as the electronic band structures, DOS and the electron effective masses for wurtzite ZnO and Mg$_x$Zn$_{1-x}$O with different Mg mole fractions have calculated. Figure 2 shows the conduction band minimum (CBM) and valance band maximum (VBM) values for the ZnO and Mg$_x$Zn$_{1-x}$O structures for different Mg mole fractions. Figure 3 shows the DOS of these structures. It is clear that the Mg$_x$Zn$_{1-x}$O structures have a wide direct band. The band gap of
Figure 2. The CBM and VBM of wurtzite $\text{Mg}_x\text{Zn}_{1-x}\text{O}$, for different Mg mole fractions (0–31.25)%.

Figure 3. DOS of wurtzite $\text{Mg}_x\text{Zn}_{1-x}\text{O}$, for different Mg mole fractions 0–31.25%.
pure ZnO is 3.35 eV, which is very close to the experimental value of ZnO [26]. The obtained band gap values of wurtzite Mg$_x$Zn$_{1-x}$O with different Mg mole fractions are listed in Table 1. In order to compare the band gap energies of this study with other experimental band gap results of wurtzite Mg$_x$Zn$_{1-x}$O, a comparison between the experimental and our results is shown in Figure 4. The calculated band gap energies of Mg$_x$Zn$_{1-x}$O show good consistency with the experimental results given in the literature [25,27–29].

The band gap values and the Mg mole fraction dependent behaviour show important similarity with the Ashrafi and Segawa’s study [25], where the experimental lattice parameter’s are taken. Because when ZnO is alloyed with MgO, an Mg$_x$Zn$_{1-x}$O structure can be obtained which has a stable wurtzite crystal

<table>
<thead>
<tr>
<th>Mg mole fraction ($x$) (%)</th>
<th>Band gap energy (eV)</th>
<th>Electron effective mass ($m_e^*$)</th>
<th>Static dielectric constant ($\varepsilon_0$)</th>
<th>High-frequency dielectric constant ($\varepsilon_{\infty}$)</th>
</tr>
</thead>
<tbody>
<tr>
<td>0</td>
<td>3.35</td>
<td>0.383</td>
<td>3.06</td>
<td>3.14</td>
</tr>
<tr>
<td>6.25</td>
<td>3.48</td>
<td>0.390</td>
<td>3.01</td>
<td>2.80</td>
</tr>
<tr>
<td>12.5</td>
<td>3.59</td>
<td>0.396</td>
<td>2.97</td>
<td>2.70</td>
</tr>
<tr>
<td>18.75</td>
<td>3.70</td>
<td>0.404</td>
<td>2.92</td>
<td>3.23</td>
</tr>
<tr>
<td>25</td>
<td>3.79</td>
<td>0.412</td>
<td>2.86</td>
<td>3.43</td>
</tr>
<tr>
<td>31.25</td>
<td>3.86</td>
<td>0.416</td>
<td>2.81</td>
<td>3.74</td>
</tr>
</tbody>
</table>

Figure 4. The calculated band gap values of Mg$_x$Zn$_{1-x}$O structures as a function of Mg mole fractions and the literature values. The solid line is drawn as a guide to the eye.
structure up to 30–36% of Mg mole fraction as pointed before. Further increase of the Mg composition resulted in the inclusion of rocksalt structure in the total structure. This situation was overcome by the use of a short-period MgO/ZnO superlattice instead of the required wurtzite MgZnO alloy. This process is called quasi-ternary alloying [30]. However, it is out of the scope of this article, therefore upto 35%, Mg mole fraction results are shown in Figure 4.

Figure 5 shows the electron effective masses of Mg$_{x}$Zn$_{1-x}$O structures as a function of the Mg mole fractions. The calculated electron effective masses of the studied structures increase linearly with increasing Mg mole fractions similar to the results found in the literature [31,32]. Although the effective mass values of this study are higher than other theoretical results and very close to the experimental results.

The static dielectric constants and the high-frequency dielectric constants of wurtzite Mg$_{x}$Zn$_{1-x}$O in the c-axis direction are calculated and listed in Table 1. The static dielectric constants decrease with the increasing Mg mole fractions. For the pure ZnO, the experimental static dielectric constant value is found to be between 8.36 and 8.91 [33]. Our calculations underestimate the static dielectric constant values. Ab initio results of the static dielectric constant of wz-ZnO are known to be underestimated with respect to the experimental results [34,35]. In this study, similar behaviour for the wz-MgZnO is observed. The high-frequency dielectric constants of Mg$_{x}$Zn$_{1-x}$O are very consistent with the experimental results [36].
Figure 6. The real and the imaginary parts of the dielectric function of wurtzite Mg$_x$Zn$_{1-x}$O as a function of photon energy for (a) perpendicular direction to the c axis ($E \perp c$) and (b) parallel direction to the c axis ($E || c$).

Figure 6 shows the real and the imaginary parts of the dielectric function for (a and c) perpendicular direction to the c axis ($E \perp c$) and (b and d) parallel direction to the c axis ($E || c$) as a function of photon energy. The curves and the peaks move towards the higher energies (the lower wavelengths) as the Mg mole fraction increases.

Figure 7. The absorption spectrum of wurtzite Mg$_x$Zn$_{1-x}$O as a function of photon energy for (a) perpendicular direction to the c axis ($E \perp c$) and (b) parallel direction to the c axis ($E || c$).

Figure 7. The absorption spectrum of wurtzite Mg$_x$Zn$_{1-x}$O as a function of photon energy for (a) perpendicular direction to the c axis ($E \perp c$) and (b) parallel direction to the c axis ($E || c$).
Figures 7 and 8 shows the absorption and transmittance spectrums of wurtzite Mg$_x$Zn$_{1-x}$O in (a) the perpendicular direction to the c axis ($E \perp c$) and (b) in the parallel direction to the c axis ($E || c$) as a function of the photon energies, respectively. It is clear that the absorption edges of the Mg$_x$Zn$_{1-x}$O shift towards the higher energies (i.e. towards the lower wavelengths) as the Mg mole fractions increase. Similar to the absorption spectrum, the transmittance spectrum of wurtzite Mg$_x$Zn$_{1-x}$O shifted to higher energy with increasing Mg mole fraction in the crystal. In addition, Figure 8 shows the anisotropic optical properties of wurtzite Mg$_x$Zn$_{1-x}$O due to the asymmetry in its crystal structure.

4. Conclusion

In this study, the electronic and the optical properties of wurtzite Mg$_x$Zn$_{1-x}$O for different Mg mole fractions have been calculated by DFT in which the electronic interactions are described within the GGA + $U$ formalism. The results show that the electronic band gap energy values of the Mg$_x$Zn$_{1-x}$O increase gradually with the increasing Mg mole fraction. The results are highly consistent with the experimental band gap results of wurtzite Mg$_x$Zn$_{1-x}$O. In addition, the calculated electron effective masses of Mg$_x$Zn$_{1-x}$O shows a linear increase with increasing Mg mole fraction. The static dielectric constants decrease with the
increasing Mg mole fractions. The high-frequency dielectric constants of Mg$_x$Zn$_{1-x}$O are very close to the experimental results in the literature. Furthermore, the absorption edges and the transmittance spectra of Mg$_x$Zn$_{1-x}$O structures are shifted to higher energy values with increasing Mg mole fraction as known in the literature. The results of this study showed that GGA + U formalism with analytical potential geometric optimisation and experimental lattice parameters usage had result in better accuracy in obtaining the correct electronic and optical properties of ZnO and MgZnO. This method can be suggested to be used in other wide-bandgap oxide semiconductors, including BeZnO, CdZnO and their quaternary compounds.

**Disclosure statement**

No potential conflict of interest was reported by the authors.

**Funding**

This work was supported by ‘Turkish Scholarships Fund’ and TUBITAK under Project No. 116F197. E.O. acknowledges partial support from the Turkish Academy of Sciences. S.B.L was supported in part by the Distinguished Young Scientist Award of Turkish Academy of Sciences [TUBA-GEIP 2016].

**References**


